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Synthesis, Structural, and Adsorption Properties and Thermal Stability of Nanohydroxyapatite/Polysaccharide Composites

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Abstract

A series of composites based on nanohydroxyapatite (nHAp) and natural polysaccharides (PS) (nHAp/agar, nHAp/chitosan, nHAp/pectin FB300, nHAp/pectin APA103, nHAp/sodium alginate) was synthesized by liquid-phase two-step method and characterized using nitrogen adsorption–desorption, DSC, TG, FTIR spectroscopy, and SEM. The analysis of nitrogen adsorption–desorption data shows that composites with a nHAp: PS ratio of 4:1 exhibit a sufficiently high specific surface area from 49 to 82 m²/g. The incremental pore size distributions indicate mainly mesoporosity. The composites with the component ratio 1:1 preferably form a film-like structure, and the value of S_{BET} varies from 0.3 to 43 m²/g depending on the nature of a polysaccharide. Adsorption of Sr(II) on the composites from the aqueous solutions has been studied. The thermal properties of polysaccharides alone and in nHAp/PS show the influence of nHAp, since there is a shift of characteristic DSC and DTG peaks. FTIR spectroscopy data confirm the presence of functional groups typical for nHAp as well as polysaccharides in composites. Structure and morphological characteristics of the composites are strongly dependent on the ratio of components, since nHAp/PS at 4:1 have relatively large S_{BET} values and a good ability to adsorb metal ions. The comparison of the adsorption capacity with respect to Sr(II) of nHAp, polysaccharides, and composites shows that it of the latter is higher than that of nHAp (per 1 m² of surface).

Keywords: Nanohydroxyapatite, High- and low-esterified pectins, Agar, Sodium alginate, Chitosan, Composites, Sr(II) adsorption

Background

In recent years, intense researches are carried out to prepare bio-hydroxyapatite composites with desired biological, physical, and mechanical properties. Hydroxyapatite and its composites are of interest due to applications in medicine. The physicochemical properties and biocompatibility make them a very attractive object for investigations both in vivo and in vitro [1–19].

Modification of hydroxyapatite (HAp) with such natural polysaccharides as chitosan [3–6], sodium alginate [7–13], agar [14, 15] and pectins [16–19], or embedding HAp nanoparticles into a polymer matrix as a filler allows one to control the morphological, structural, and

mechanical properties of composites to enhance the functional use. The composites based on chitosan with HAp or alginate are mainly used to treat bone implants [3–8, 11–14, 16–18] or to use as drugs carriers [9, 10, 15].

Analysis of the literature shows that the synthesis of HAp/PS composites, their use, and control of the properties are far from exhausted ones. The use of HAp/PS composites as adsorbents could be promising since the components alone show a high adsorption capacity with respect to heavy metal cations [20–34]. Creation of composites allows one to control the structure of the materials to improve the morphology and to enhance the adsorption properties. It is known that natural polysaccharides are good sorbents of some kinds of dyes [29–31, 35] and heavy metal ions [23–33] because of specific interactions of the amino and hydroxyl groups with adsorbates [37]. The amino groups of polysaccharides can be cationized

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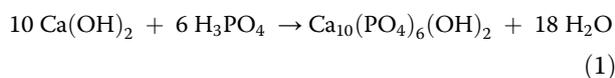
that allows the effective adsorption of ionic dyes [38]. However, the use of some polysaccharides in their native form is difficult since the viscosity of the solutions is high even at a low concentration because of tendency to gelling. Therefore, composites with PS can be more appropriate for the adsorption applications due to diminution of the mentioned negative effects [39–41]. Immobilization of macromolecules on a HAp surface allows an increase in sorption activity of the composites compared to the components alone.

The objective of this work was the synthesis of nHAp/polysaccharide composites and the study of the structural and morphological characteristics, the thermal behavior, and sorption capacity with respect to Sr(II).

Methods

Materials

The nHAp/PS composites were synthesized by mixing of nHAp suspensions with polysaccharides solutions in two stages. The first stage was the synthesis of nHAp by a wet chemical method. In the reaction



calcium hydroxide (Aldrich) and phosphoric acid (POCh, Gliwice) were used as 1 M aqueous solutions of 0.18 and 0.3 L, respectively. The H_3PO_4 solution was dropped into the Ca(OH)_2 suspension placed in a flask for 15 min. While dropping the reaction, mixture was stirred vigorously and then dried in a dryer at 80 °C for 24 h. A white sediment with crystalline hydroxyapatite was obtained. Then the sediment was washed with redistilled water till the constant value of redistilled water conductivity was achieved. The average crystallite size determined from XRD patterns using Scherrer's equation applied to a peak at $2\theta = 25.9$ was 26 nm. The degree of crystallinity [42] was 22%.

The second stage was the synthesis of nHAp/polysaccharide composites using chitosan (deacetylation degree 82%, "Bioprogress" CJSC, Moscow, Russia), high-etherified apple pectin APA 103 at the degree of etherification (DE) of 66–68% and low-etherified apple pectin APA 300 FB with galacturonic acid with free carboxyl groups 64–69% (Andre Pectin, China), and sodium alginate (SA) at a mass fraction of the basic substance of 99.0% (China) as received. nHAp composites were prepared by mixing of the nHAp suspension and PS solution. Additionally, the polysaccharide solution (2 wt.%) and nHAp suspension (4 wt.%) sonicated for 3 min were prepared using distilled water, mixed at the nHAp/PS ratio of 1:1 and 4:1, and stirred for 30 min. Then the nHAp/PS suspensions were dried at 40 °C in air.

The hydroxyl groups are the main functional groups of PS, which can be esterified or oxidized. The carboxyl groups of uronic acid can be esterified, and the amino groups of amino sugars can be acylated. Modified PS are capable to create strong complexes with metal ions, as well as with polar low-molecular weight organics.

The formation of composites occurs due to strong interactions of the phosphate and hydroxyl groups of nHAp with the COO^- , OH, and other polar groups in PS [18]. The polysaccharide molecules also tend to form the hydrogen bonds with each other resulting in gelation of their aqueous solutions upon heating at certain temperatures. The calcium phosphate ions can be trapped in the PS chains. The cross-linking reactions may occur in the composites. Therefore, nHAp nanoparticles could be well distributed in the PS network and remained in stable state for a long period.

Fourier Transform Infrared (FTIR) Spectroscopy

FTIR spectra of powdered samples (grinded with dry KBr at the mass ratio 1:9) over the 4000–400 cm^{-1} range were recorded using a ThermoNicolet FTIR spectrometer with a diffuse reflectance mode.

Scanning electron microscopy (SEM)

The surface morphology of composites was analyzed using field emission scanning electron microscopy employing a QuantaTM 3D FEG (FEI, USA) apparatus operating at the voltage of 30 kV.

Textural characteristics

Specific surface areas and pore volumes were determined from low-temperature nitrogen adsorption isotherms using a Micromeritics ASAP 2020 or 2405N adsorption analyzer. Before measurements, the samples were outgassed at 80 °C for 12 h. The nitrogen desorption data were used to compute the pore size distributions (PSD, differential $f_V \sim dV_p/dR$ and $f_S \sim dS/dR$) using a self-consistent regularization (SCR) procedure under non-negativity condition ($f_V \geq 0$ at any pore radius R) at a fixed regularization parameter $\alpha = 0.01$ using a model of voids (V) between spherical nonporous nanoparticles packed in random aggregates (V/SCR model) [43]. The differential PSD with respect to the pore volume $f_V \sim dV/dR$, $\int f_V dR \sim V_p$ were re-calculated to incremental PSD (IPSD) at $\Phi_V(R_i) = (f_V(R_{i+1}) + f_V(R_i))(R_{i+1} - R_i)/2$ at $\sum \Phi_V(R_i) = V_p$. The f_V and f_S functions were also used to calculate contributions of micropores (V_{micro} and S_{micro} at $0.35 \text{ nm} < R < 1 \text{ nm}$), mesopores (V_{meso} and S_{meso} at $1 \text{ nm} < R < 25 \text{ nm}$), and macropores (V_{macro} and S_{macro} at $25 \text{ nm} < R < 100 \text{ nm}$).

Thermal analysis

Thermal analysis was carried out using a STA 449 Jupiter F1 (Netzsch, Germany) apparatus, sample mass $\sim 16 \text{ mg}$

placed into a corundum crucible, air flow of 50 mL min⁻¹, a heating rate of 10 °C min⁻¹, temperature range of 30–950 °C, and S TG-DSC sensor thermocouple type. Empty corundum crucible was used as a reference. The gaseous products emitted during decomposition of the materials were analyzed by using a FTIR Bruker (Germany) spectrometer and QMS 403D Aëolos (Germany) coupling on-line to STA instrument. The QMS data were gathered in the range from 10 to 200 a.m.u. The FTIR spectra were recorded in the range of 4000–600 cm⁻¹ with 16 scans per spectrum at a resolution of 4 cm⁻¹.

Adsorption of Sr(II)

The adsorption of Sr(II) ions vs. pH at the composite/electrolyte solution interface was determined by the means of the radioisotope method. The initial concentration of Sr(II) ions was 10⁻⁴ M. NaCl (0.001 mol/dm³) was used as a background electrolyte, and pH was changed from 3 to 10. The adsorption measurements were complemented by the potentiometric titration of the composites in the suspensions and the electrophoresis measurements. The adsorption measurements were performed in a thermostated Teflon vessel at 25 °C. To eliminate CO₂, all the potentiometric measurements and adsorption experiments were carried out under the nitrogen atmosphere. The pH values were measured using a set of glass REF 451 and calomel pHG201-8 electrodes with Radiometer assembly. Radioactivity of the solutions before and after adsorption was measured using a LS 5000 TD Beckmann liquid scintillation counter. Because ⁹⁰Sr decays to the radioactive ⁹⁰Y, the measurements were carried out in two channels in order to calculate radioactivity of ⁹⁰Sr.

Results and Discussion

Textural Characterization

The BET surface area and pore volume of composites (Table 1) depend on the content and type of PS. The

initial nHAp has S_{BET} of 105 m²/g and V_p of 0.54 cm³/g, while for composites, they decrease with increasing PS concentration due to filling of inter-particle voids in aggregates by polymer molecules. The shape of the nitrogen adsorption–desorption isotherms (Fig. 1) corresponds to type II with hysteresis loop H3 of the IUPAC classification [44, 45] corresponding to the textural porosity of aggregates of nonporous nanoparticles.

The hysteresis loop shape indicates dominant contribution of mesopores (filled by adsorbed nitrogen during the measurements). It should be noted that in the case of highly disperse materials, only a certain part of pores can be filled by nitrogen because large macropores remain empty, i.e., $V_p < V_{\text{em}} = 1/\rho_b - 1/\rho_0$, where ρ_b and ρ_0 are bulk and true densities of the materials.

The pore size distribution functions (Fig. 2) confirm the conclusion based on the isotherm shape (Fig. 1) that the composites are mainly mesoporous, since contributions of micropores and macropores are small (Table 1). The first peak of the PSD corresponds to narrow voids between nanoparticles/polymers closely located in the same aggregates. Broader voids can be between neighboring aggregates. The PSD show that different PS form different shells of nanoparticles, especially in the range of narrow pores at $R < 10$ nm (Fig. 2). Therefore, the average pore radius $R_{p,v}$ in nHAp/PS at the ratio of 4:1 is not the same, and it is in the range of 13.9–17.0 nm corresponding to mesopores (Table 1). Despite filling of voids by PS, the values of $R_{p,v}$ increase with increasing PS content. These changes can be explained by several factors. First, narrow voids are more strongly filled by PS than broad voids. Second, adsorption of PS results in compacting of aggregates of nanoparticles and agglomerates of aggregates (see Figs. 1, 2, 3, and 4, Table 1).

A film-like, near-monolithic structure is formed in nHAp/PS at the ratio of 1:1 (Fig. 3b, d) as evidenced by low values of the specific surface area. However, some

Table 1 Textural characteristics of initial nHAp and nHAp/PS nanocomposites

| Sample | S_{BET} (m ² /g) | S_{micro} (m ² /g) | S_{meso} (m ² /g) | S_{macro} (m ² /g) | V_p (cm ³ /g) | V_{micro} (cm ³ /g) | V_{meso} (cm ³ /g) | V_{macro} (cm ³ /g) | $R_{p,v}$ (nm) |
|-------------------|--------------------------------------|--|---------------------------------------|--|----------------------------|---|--|---|----------------|
| nHAp | 106 | 7.3 | 85 | 14 | 0.54 | 0.004 | 0.35 | 0.19 | 22.3 |
| nHAp/agar 4:1 | 75 | 1.2 | 71 | 2.9 | 0.35 | 0.001 | 0.31 | 0.05 | 17.0 |
| nHAp/agar 1:1 | 43 | 14 | 29 | 0.1 | 0.26 | 0.009 | 0.24 | 0.05 | 11.3 |
| nHAp/SA 4:1 | 82 | 6.4 | 73 | 2.6 | 0.42 | 0.002 | 0.36 | 0.05 | 16.7 |
| nHAp/SA 1:1 | 1 | 0.0 | 0.8 | 0.2 | 0.10 | 0.0 | 0.08 | 0.02 | 63.6 |
| nHAp/chitosan 4:1 | 53 | 0.1 | 52 | 1.3 | 0.25 | 0.0 | 0.23 | 0.02 | 15.4 |
| nHAp/chitosan 1:1 | 8 | 0.3 | 6.2 | 1.2 | 0.11 | 0.0 | 0.03 | 0.08 | 63.4 |
| nHAp/FB300 4:1 | 49 | 7.4 | 41 | 0.9 | 0.25 | 0.003 | 0.23 | 0.02 | 14.7 |
| nHAp/FB300 1:1 | 0.3 | 0.1 | 0.0 | 0.2 | 0.00 | 0.001 | 0.00 | 0.001 | 88.6 |
| nHAp/APA103 4:1 | 56 | 0.4 | 55 | 1.1 | 0.25 | 0.0 | 0.23 | 0.02 | 13.9 |

Specific surface area in total (S_{BET}), micropores (S_{micro}), mesopores (S_{meso}), and macropores (S_{macro}) and respective pore volumes (V_p , V_{micro} , V_{meso} , V_{macro}). $R_{p,v}$ is the average pore radius with respect to the pore volume

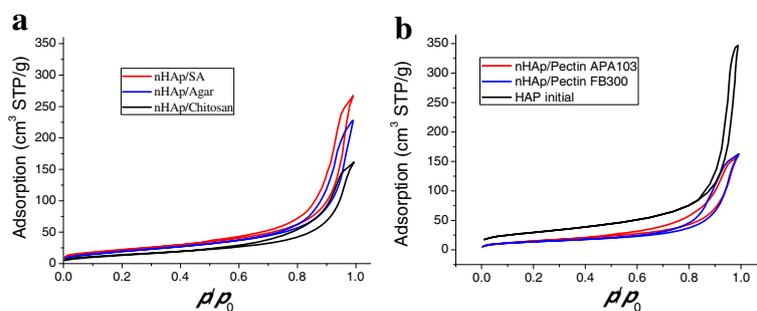


Fig. 1 Nitrogen adsorption-desorption isotherms for composites with nHAp:PS ratio 4:1

porosity of the composites remains. At the component ratio of 4:1, the structure of composites is more porous and can be described as multimodal aggregates of nHAp/PS with sizes over a wide range of 20–250 nm (Fig. 3c, e). Similar structure of aggregates of primary particles inherent to initial nHAp (Fig. 3a), and it remains for composites nHAp/pectins and nHAp/SA (Fig. 4). Thus, a relatively high value of S_{BET} and porous structure of these composites (Table 1, Figs. 1, 2, 3, and 4) indicate the prospects for their use as better adsorbents than those at 50 wt.% of PS.

Fourier Transform Infrared Spectroscopy (FTIR)

The IR spectrum of hydroxyapatite (Fig. 5) exhibits characteristic bands at 561 and 602 cm^{-1} corresponding to triply degenerated bending modes of the O–P–O bond vibrations in the phosphate groups [46–49]. A band at 472 cm^{-1} corresponds to double-degenerated bending modes of the O–P–O bonds [46–48]. A band at 962 cm^{-1} can be attributed to non-degenerated symmetric stretching modes of the P–O bonds [46–50]. Bands at 1032 and 1101 cm^{-1} are due to triply degenerated asymmetric stretching vibrations of the P–O bonds.

The presence of $(\text{CO}_3)_2$ groups is confirmed by bands at 1414 cm^{-1} , which are assigned to stretching mode of the $(\text{CO}_3)_2$ groups [46, 51–53], and at 1465 cm^{-1} , which can be attributed to the stretching modes of the $(\text{CO}_3)_2$ groups in A-type of carbonated apatite [46, 54, 55].

OH groups and adsorbed water molecules, presented in the hydroxyapatite sample, give peaks at 875 cm^{-1} (hydrogen phosphate groups) and 2856–3656 cm^{-1} with a maximum at 3400 cm^{-1} .

All characteristic bands of hydroxyapatite remain in the IR spectra of the nHAp/PS composites (Fig. 5), but their intensity decreases with decreasing content of hydroxyapatite. The appearance of the bands at 2927 and 2851 cm^{-1} is due to C–H asymmetric and symmetric stretching vibrations in the aliphatic CH_2 groups of PS. Especially noticeable increase in intensity of broad band with a maximum at 3445 cm^{-1} is due to the OH groups in PS forming the hydrogen bonds with each other or adsorbed water molecules. Additionally, the N–H bonds in the amino groups of chitosan give the bands at $\sim 3350 \text{ cm}^{-1}$. Similar features of the IR spectra of nHAp/pectin are shown in Fig. 5b. Pectin molecules include few hundred linked galacturonic acid residues forming a long molecular chain with polygalacturonic acid subunits is methoxylated. The pectin molecules contain a large amount of carboxyl (free and esterified), hydroxyl, methoxyl, and acetyl groups. The bands at 2926 and 2852 cm^{-1} of the C–H stretching vibrations and a broad band with a maximum at 3445 cm^{-1} increase with increasing content of pectins. The IR spectra of pectins are characterized by bands at 1750–1700 cm^{-1} related to the stretching vibrations of the carbonyl, ester, and carboxyl groups. The IR spectra of nHAp/sodium alginate

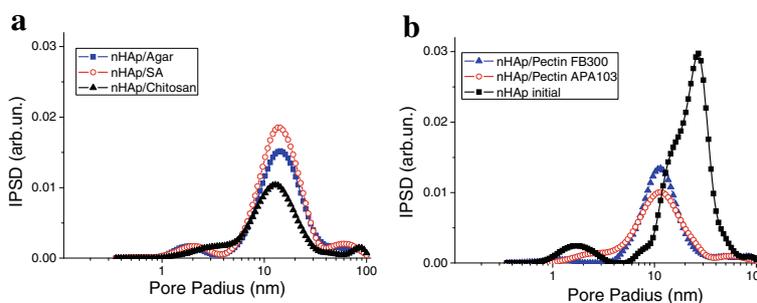


Fig. 2 Incremental pore size distributions for composites with ratio nHAp:PS 4:1

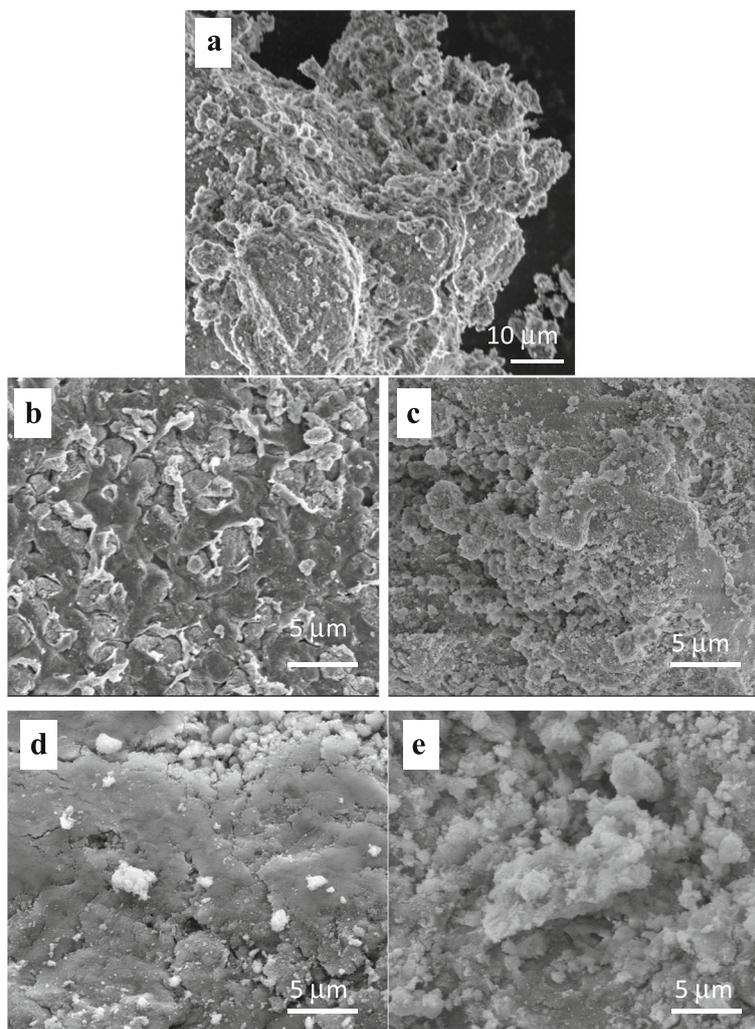


Fig. 3 SEM images of initial nHAp (a), nHAp/chitosan (b) 1:1 and (c) 1:4 and nHAp/agar (d) 1:1 and (e) 1:4

(Fig. 5c) show similar bands of the hydroxyl, ether, and carboxylic groups, as well as the O–H and C–H stretching vibrations of alginate. Bands at 1633 and 1460 cm^{-1} can be attributed to the asymmetric and symmetric stretching vibrations of carboxylate salt ions. These bands can be used to characterize structures of alginate, its derivatives, and ingredients.

Thermal Analysis

The thermal characteristics (TG, DTG, and DSC) of nHAp/PS, nHAp, and polysaccharides were studied upon heating of samples in air (Figs. 6 and 7, Additional file 1: Table S1). Our previous studies [56] have shown that in case of thermal decomposition of hydroxyapatite (Fig. 6d), the weight losses are results of the process of desorption of physically adsorbed water and dehydroxylation in temperature range to 200 °C and removing of carbonates at higher temperatures. According to

literature [57–59], the thermal decomposition of organic molecules is very complicated and occurs in a few main stages. The first stage comprises physicochemical transformation (dehydration, melting, changes in conformation of molecules, initial defragmentation etc.) and occurs at low temperature. The processes of defragmentation and partial oxidation of the H atoms prevail mainly in temperature range to 400 °C. In the range above 500 °C, the peaks on DTG or DSC curves are due to processes thermo-oxidation of the H and N atoms and pyrolysis of charcoal.

The low-temperature mass loss from 30 to 150 °C for PS and nHAp/PS corresponds to intact water desorption. The main weight loss was found for the PS degradation step (150–350 °C) [60, 61]. This step decomposition of organic molecules was confirmed by the increasing peaks for water (m/z 18) and carbon dioxide (m/z 44) in the mass spectra (Fig. 8) of analyzed samples. In nHAp/PS,

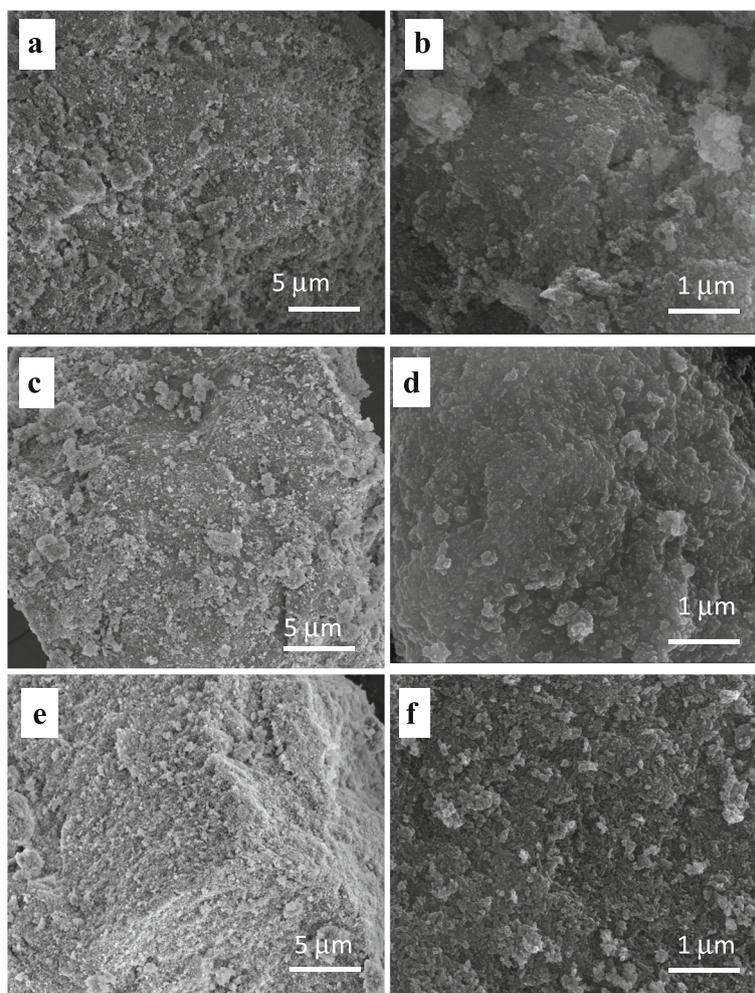


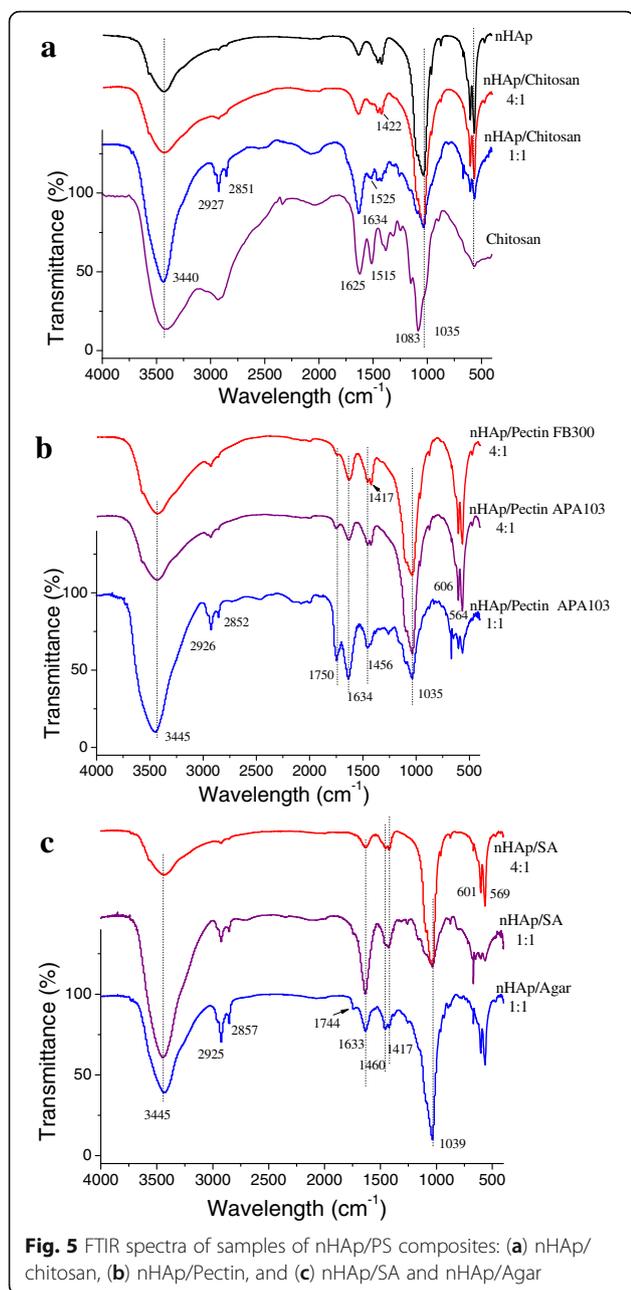
Fig. 4 SEM images of nHAp/pectin FB 300 (a, b), nHAp/pectin APA103 (c, d), nHAp/SA (e, f) components ratio 1:4

condensation and elimination of hydroxyl groups occur at 150–250 °C [39]. TG and DTG curves of chitosan alone demonstrate the polymer chain decomposition from 197 to 276 °C with maxima at 211.8 and 237.6 °C. For composites nHAp/chitosan, only a single peak is observed with a maximum at 234.5 °C (Fig. 6a, Additional file 1: Table S1). This difference can be attributed to the changes in the structure or conformation of individual and adsorbed chitosan. TG and DTG curves of sodium alginate are characterized by decomposition of the polymer chain from 210 to 368.7 °C with maxima at 246.2 and 350.1 °C, which are most likely caused by condensation of hydroxyl groups and destruction of the organic component [62, 63]. For nHAp/sodium alginate, temperatures of peaks correspond to PS decomposition slightly shifted toward lower temperatures. This indicates some decrease in thermal stability of sodium alginate in the composite compared to sodium alginate alone. Decomposition of the polymer chain of agar occurs from 243 to 384 °C with a maximum at 297.1 °C. For

nHAp/agar, the peak position corresponding to PS degradation does not practically change, but the width of the peak decreases.

TG and DTG curves of pectin FB300 characterize decomposition of the polymer chain in the first stage from 203 to 337 °C with maxima at 226.3 and 302.9 °C (Fig. 6c, Additional file 1: Table S1). For composites, three degradation peaks at 204.1, 250.1, and 316.2 °C are observed (Fig. 6d, Additional file 1: Table S1). The temperature range of PS degradation becomes wider compared with the pectin alone. The amount of physically adsorbed water is less in the composite than the pectin alone. Decomposition of pectin APA103 begins at lower temperatures than that of pectin FB300. The peak of pectin degradation in composites is slightly shifted toward lower temperatures.

Thermal effects upon degradation of polysaccharides can be estimated from the DSC data (Fig. 7). A weak endothermic peak between 50 and 150 °C with a maximum of ca. One hundred degree Celcius can be attributed to



desorption of intact water. Thermodegradation of polysaccharides is usually accompanied by an exothermic effect. Typical DSC curves (Fig. 7) show three main peaks upon thermal analysis of sodium alginate and chitosan, and two peaks for agar. It is noteworthy that the intensive peaks distinguishable for initial (bulk) sodium alginate on DSC curve at 583.6 °C (Fig.7a) and DTG curve at 580.7 °C (Fig.6a) strongly changed for composite nHAp/SA: DSC peak disappears (Fig.7b), and DTG peak has much smaller intensity and shifted to temperature 672.7 °C (Fig.6b). The similar regularities are also observed for other composites HAp/PS: high-temperature peaks distinguishable on the DSC curves

for bulk agar at 460.7 °C (Fig.7a), for bulk pectin FB300 at 746.7 and 586.6 °C, for bulk pectin APA103 at 593.3 °C (Fig.7c) are not observed on DSC curves for the corresponding composites nHAp/PS (Fig.7b, d). In the case of chitosan, all temperature peaks on DCS curve of bulk polysaccharide appear in DCS curve for the nHAp/chitosan composite but shifted to lower temperatures (Fig.7a, b). On DTG curves of composites agar/MS and pectin FB300/PS, the shift toward lower temperatures is observed for high-temperature peaks compared to the DTG curves for the initial polysaccharides: DTG peak for bulk agar at 457.5 °C shifted to 381.2 °C; peak on the DTG curve for pectin FB300 at 742.7 °C has disappeared on DTG curve for the composite; and the peak at 581.3 °C shifted to 504.4 °C. DTG peaks shifted slightly for nHAp/APA103 composites compared with initial pectin (Fig. 6c, d). Such peculiarities show that a strong interaction PS with nHAp results in a significant change in the thermal properties of PS. The multiple exothermic peaks of pectin degradation are observed at $T > 150$ °C, wherein for pectin FB300, it is not clear separation of peaks indicating the complexity and manifold of processes of degradation, while for pectin APA103 main peak is at 436 °C.

Adsorption of Sr(II)

Sr^{2+} ions occur in a non-hydrolyzed form in the aqueous solutions up to pH 10.5, since it does not form sparsely soluble oxides or hydroxides. Thus, Sr^{2+} ions are convenient to study the adsorption onto a surface of composites. The study of Sr^{2+} adsorption on HAp was described in detail previously [20]. The Sr^{2+} adsorption involves the surface hydroxyls of nHAp according to ion-change mechanism:



The pH of solution is an important parameter that controls adsorption process because of ionization of surface functional groups and alteration of the solution composition. Figure 9 shows the pH dependences of the Sr^{2+} adsorption from 0.0001 M solutions for the initial nHAp and nHAp/PS composites. As it can be seen, a monotonic increase in the Sr^{2+} adsorption on initial nHAp is observed with increasing pH due to peculiarities of hydrolysis on the hydroxyapatite surface (Fig. 9a). The positively charged $=\text{CaOH}_2^+$ species and neutral $=\text{POH}^0$ sites prevail in acidic solutions. Due to a high pH value, the surface of hydroxyapatite is deprotonated, releasing H^+ ions in the solution and causing a shift of pH to lower value. The negatively charged $=\text{PO}^-$ sites and neutral $=\text{CaOH}^0$ sites predominate in alkaline solutions [64]. Adsorption of Sr^{2+} ions on the hydroxyapatite surface can proceed through the exchange of Ca^{2+} ions according to the reaction [20]:

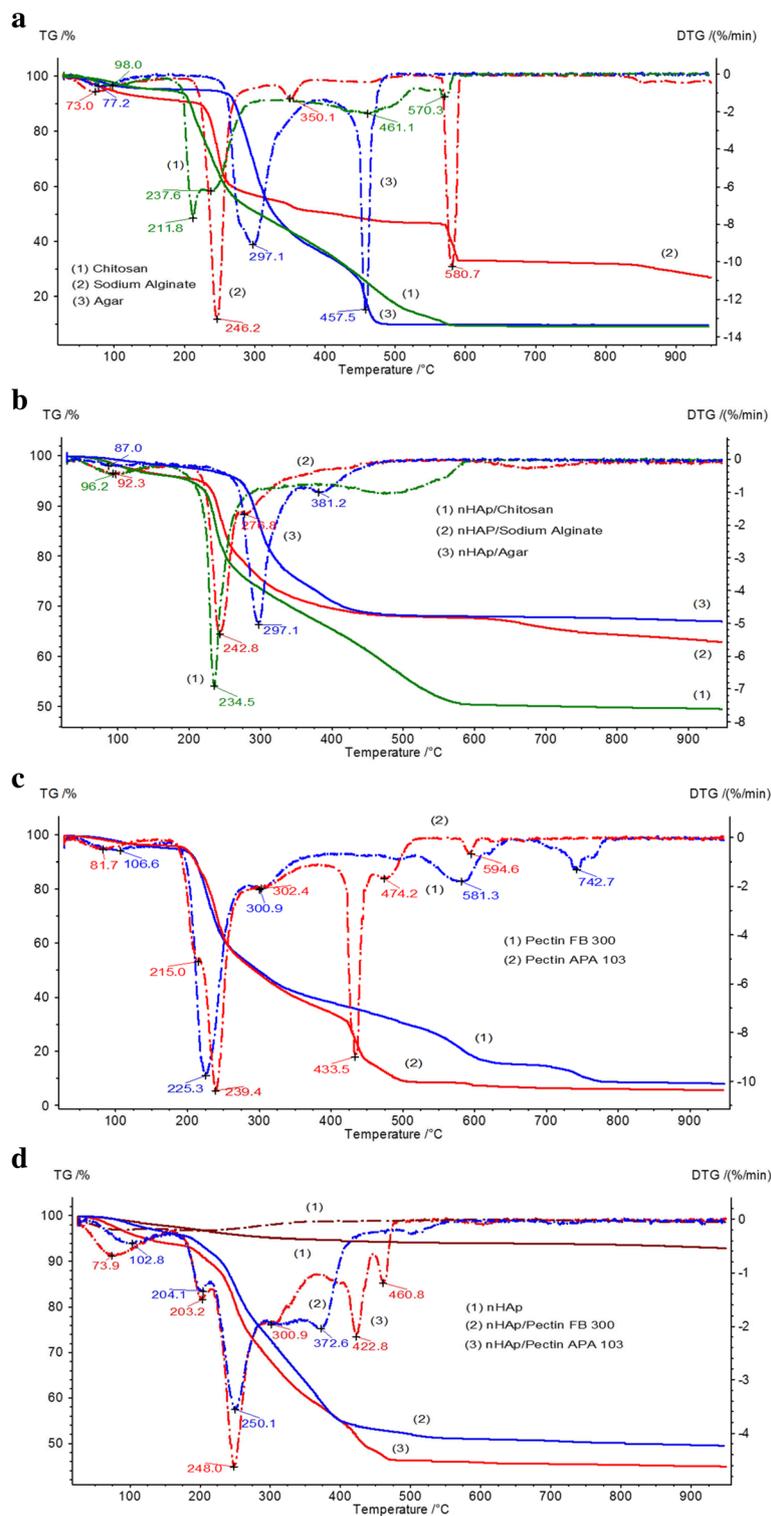


Fig. 6 DTG (broken curve) and TG (continuous curve) for initial PS and nHAp/PS composites. **a** Initial chitosan, sodium alginate, and agar. **b** nHAp/Chitosan, nHAp/SA, and nHAp/Agar composites. **c** Pectin FB300 and pectin APA103. **d** nHAp/pectin FB300 and nHAp/pectin APA103 composites

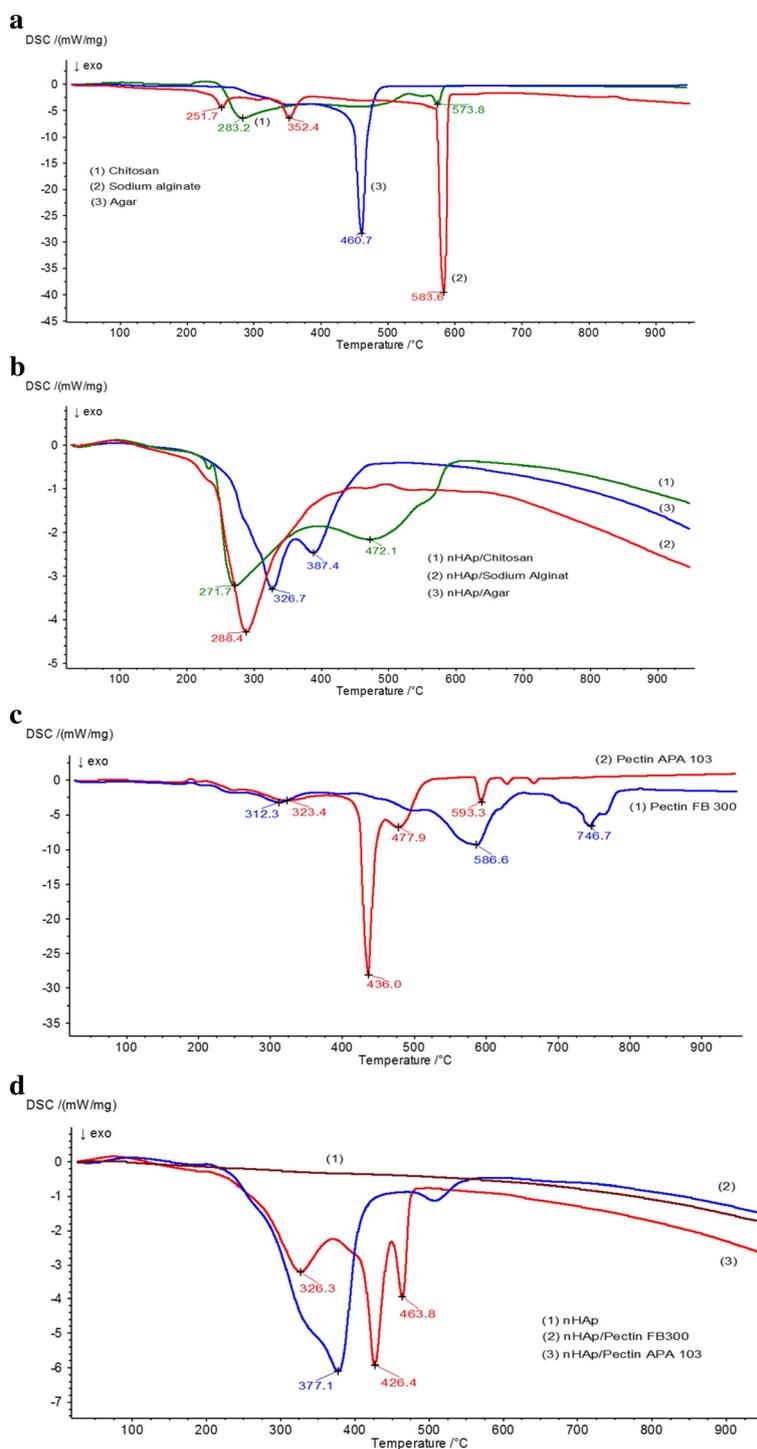


Fig. 7 DSC curves for initial PS and nHAp/PS composites. **a** Initial chitosan, sodium alginate, and agar. **b** nHAp/chitosan, nHAp/SA, and nHAp/agar composites. **c** Pectin FB300 and pectin APA103. **d** nHAp/pectin FB300 and nHAp/pectin APA103 composites



The pH dependences of the Sr²⁺ adsorption vary for composites with different PS. For nHAp/agar and nHAp/chitosan, it is also observed a monotonic pH

dependence of the Sr²⁺ adsorption, but the adsorption is higher than for the initial nHAp. The high values of the adsorption and a small remnant of the solution is achieved at a pH greater than 8 (Fig. 9a). For composites containing pectins, higher values of the adsorption are

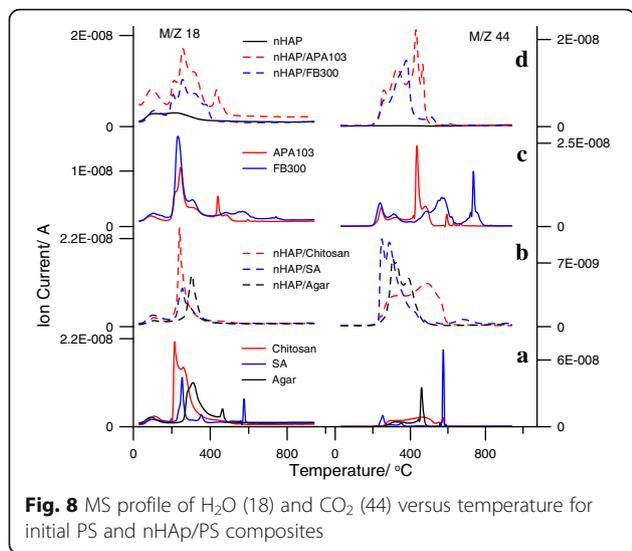


Fig. 8 MS profile of H₂O (18) and CO₂ (44) versus temperature for initial PS and nHAp/PS composites

observed in the acidic pH range. In the case of nHAp/pectin FB300 composite, the Sr²⁺ adsorption reaches 0.25 μmol/m² at pH 6.5 (Fig. 9b).

Figure 10 shows a comparison of the adsorption values and a residue concentration of Sr²⁺ ions in the solution

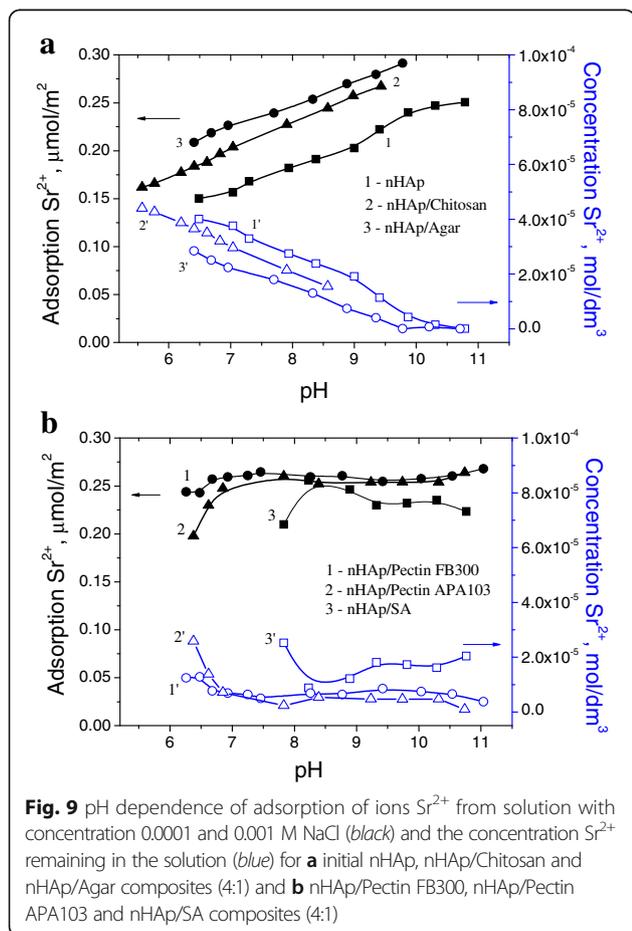


Fig. 9 pH dependence of adsorption of ions Sr²⁺ from solution with concentration 0.0001 and 0.001 M NaCl (black) and the concentration Sr²⁺ remaining in the solution (blue) for **a** initial nHAp, nHAp/Chitosan and nHAp/Agar composites (4:1) and **b** nHAp/Pectin FB300, nHAp/Pectin APA103 and nHAp/SA composites (4:1)

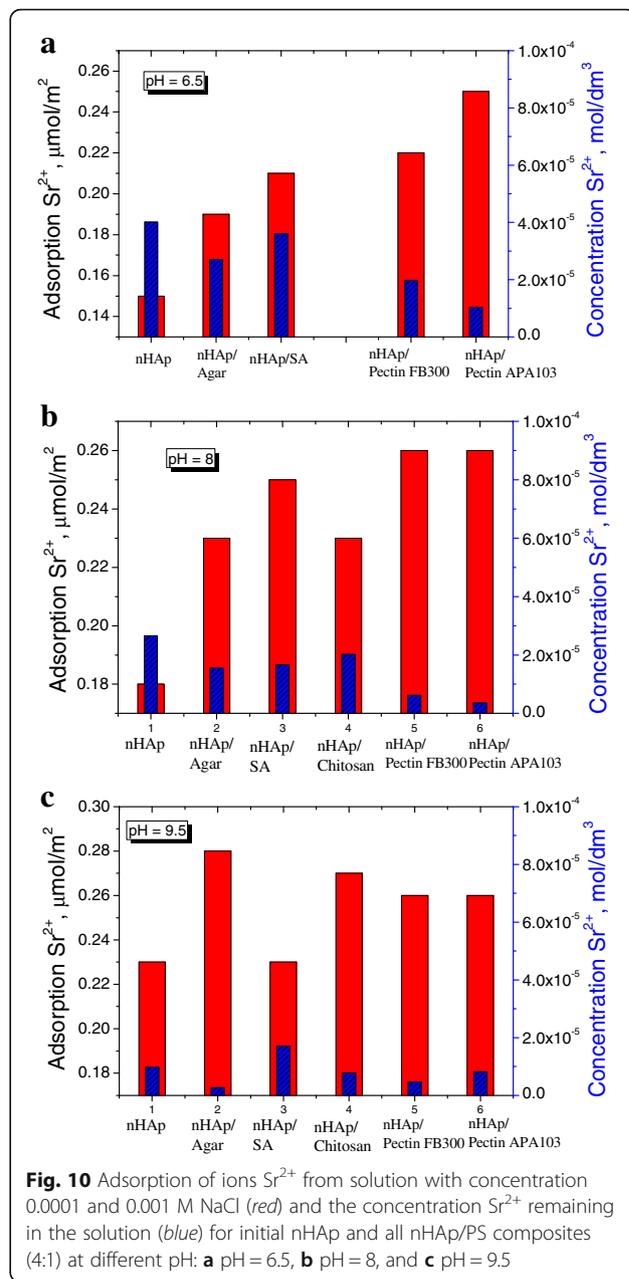


Fig. 10 Adsorption of ions Sr²⁺ from solution with concentration 0.0001 and 0.001 M NaCl (red) and the concentration Sr²⁺ remaining in the solution (blue) for initial nHAp and all nHAp/PS composites (4:1) at different pH: **a** pH = 6.5, **b** pH = 8, and **c** pH = 9.5

at pH 6.5, 8, and 9.5 for all composites. It can be seen that at pH 6.5, the composite containing pectin FB300 shows good Sr²⁺ adsorption, while at pH 8, the maximal adsorption values are inherent for nHAp/pectin FB300, nHAp/pectin APA103, and nHAp/SA. At pH 9, the composites containing chitosan and agar have the maximal adsorption.

For nHAp/PS, two mechanisms of the adsorption can be realized due to Sr(II) interactions with nHAp or polysaccharides. The adsorption of metal ions on polysaccharides occurs with participation of carboxyl groups of pectin, agar and sodium alginate, and amino groups in chitosan, which are capable of strong electrostatic interactions with metal

ions [24–26, 35–37]. Therefore, all the composites studied show the adsorption capacity with respect to Sr(II) higher than the initial nHAp.

Conclusions

Thus, nHAp/PS with different polysaccharides and different nHAp:PS ratios 4:1 and 1:1 synthesized by two-step process demonstrate certain decrease in the textural characteristics with increasing content of PS due to filling of inter-particle voids by polymers. However, the composites that have the HAP:PS ratio of 4:1 show relatively developed S_{BET} from 49 m^2/g for nHAp/pectin FB300 to 82 m^2/g for nHAp/SA. Mainly mesoporosity is the characteristic for the composites, since contribution of micro and macro pores is negligible. At the nHAp:PS ratio 1:1, a film-like structure was formed. The specific surface area and porosity largely depend on the nature of the polysaccharide, and maximal S_{BET} value of 43 m^2/g is for nHAp/agar. The thermal properties of the composites show a certain influence of nHAp on polysaccharide degradation. However, composites have sufficient thermal stability. In the composites, the polysaccharide degradation occurs at temperatures above 200 °C. It was found out that for nHAp/PS composites at the component ratio 4:1, the adsorption capacity with respect to Sr^{2+} ions is higher than for the initial nHAp. This makes these composite promising for the use as adsorbents for metal cations from aqueous media.

Additional file

Additional file 1: Table S1. Thermal behavior of studied samples (T_{onset} , T_{max} , T_{end} are an onset, maximum degradation, and end temperatures determined for DTG curves). (DOCX 20 kb)

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Authors' contributions

ES carried out the synthesis of nHAp and characterization of nanocomposites by SEM method, participated in characterization of nanocomposites by FTIR method and adsorption of Sr^{2+} . OG carried out the synthesis of nHAp/PS nanocomposites. DS participated in DSC and TG studies. KG carried out the nitrogen adsorption–desorption measurements. OG and VMG analyzed the data and drafted the manuscript. WJ and VMG revised the manuscript. All authors read and approved the final manuscript.

Competing Interests

The authors declare that they have no competing interests.

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References

- Venkatesan J, Kim SJ (2014) Nano-hydroxyapatite composite biomaterials for bone tissue engineering—a review. *J Biomed Nanotechnol* 10(10):3124–40. doi:10.1166/jbn.2014.1893
- Skwarek E, Janusz W, Gun'ko VM, Pakhlov EM, Zarko VI, Gdula K (2016) Characteristics of surface and electrochemical properties of composites with fumed metal oxides and hydroxyapatite. *Adsorption* 22:725–734. doi:10.1007/s10450-016-9770-4
- Pangon A, Saesoo S, Saengkrit N, Ruktanonchai U, Intasanta V (2016) Hydroxyapatite-hybridized chitosan/chitin whisker bionanocomposite fibers for bone tissue engineering applications. *Carbohydr Polym* 144:419–427
- Lowe B, Venkatesan J, Anil S, Shim MS, Kim SK (2016) Preparation and characterization of chitosan-natural nanohydroxyapatite-fucoidan nanocomposites for bone tissue engineering. *Int J Biol Macromol* 93:1479–1487
- Rogina A, Ivanković M, Ivanković H (2013) Preparation and characterization of nano-hydroxyapatite within chitosan matrix. *Mater Sci Eng C* 33:4539–4544
- Tsiourvas D, Sapalidis A, Papadopoulos T (2016) Hydroxyapatite/chitosan-based porous three-dimensional scaffolds with complex geometries. *Mater Today Com* 7:59–66
- Son KD, Yang DJ, Kim MS, Kang IK, Kim SY, Kim YJ (2012) Effect of alginate as polymer matrix on the characteristics of hydroxyapatite nanoparticles. *Mater Chem Phys* 132:1041–1047
- Venkatesan J, Bhatnagar I, Manivasagan P, Kang KH, Kim SK (2015) Alginate composites for bone tissue engineering: a review. *Int J Biol Macromol* 72:269–281
- Ilie A, Ghițuică C, Andronescu E, Cucuruz A, Fica A (2016) New composite materials based on alginate and hydroxyapatite as potential carriers for ascorbic acid. *Int J Pharm* 510:501–507
- Yan J, Miao Y, Tan H, Zhou T, Ling Z, Chen Y, Xing X, Hu X (2016) Injectable alginate/hydroxyapatite gel scaffold combined with gelatin microspheres for drug delivery and bone tissue engineering. *Mater Sci Eng C* 63:274–284
- Rajkumar M, Meenakshisundaram N, Rajendran V (2011) Development of nanocomposites based on hydroxyapatite/sodium alginate: synthesis and characterization. *Mater Charact* 62:469–479
- Kamalaldin NA, Yahya BH, Nurazreena A (2016) Cell evaluation on alginate/hydroxyapatite block for biomedical application. *Procedia Chemistry* 19:297–303
- Munarin F, Petrinia P, Gentilini R, Pillai RS, Dirè S, Tanzi MC, Sglavova VM (2015) Laboratorio micro- and nano-hydroxyapatite as active reinforcement for soft biocomposites. *Int J Biol Macromol* 72:199–209
- Hu J, Zhu Y, Tong H, Shen X, Chen L, Ranba J (2016) A detailed study of homogeneous agarose/hydroxyapatite nanocomposites for load-bearing bone tissue. *Int J Biol Macromol* 82:134–143
- Kolanthai E, Ganesan K, Eppele M, Kalkura SN (2016) Synthesis of nanosized hydroxyapatite/agarose powders for bonefiller and drug delivery application. *Mater Today Com* 8:31–40
- Munarin F, Petrinia P, Gentilini R, Pillai RS, Dirè S, Tanzi MC, Sglavo VM (2015) Micro- and nano-hydroxyapatite as active reinforcement for soft biocomposites. *Int J Biol Macromol* 72:199–209
- Li J, Sun H, Sun D, Yao Y, Yao F, Yao K (2011) Biomimetic multicomponent polysaccharide/nano-hydroxyapatite composites for bone tissue engineering. *Carbohydr Polym* 85:885–894
- Li J, Zhu D, Yin J, Liu Y, Yao F, Yao K (2010) Formation of nano-hydroxyapatite crystal in situ in chitosan–pectin polyelectrolyte complex network. *Mater Sci Eng C* 30:795–803
- Munarin F, Petrinia P, Barcellona G, Roversi T, Piazza L, Visai L, Tanzi MC (2014) Reactive hydroxyapatite fillers for pectin biocomposites. *Mater Sci Eng C* 45:154–161
- Janusz W, Skwarek E (2016) Study of sorption processes of strontium on the synthetic hydroxyapatite. *Adsorption*. 22(4). DOI: 10.1007/s10450-016-9761-5.
- Skwarek E, Janusz W (2016) Adsorption of Cd(II) ions at the hydroxyapatite/electrolyte solution interface. *Separ Sci Technol* 51(1):11–21

22. Skwarek E (2015) Adsorption of Cs⁺ at the hydroxyapatite/aqueous electrolyte interface. *Adsor Sci Technol* 33(6-8):575–580
23. Abdel-Halim ES, Al-Deyab SS (2011) Removal of heavy metals from their aqueous solutions through adsorption onto natural polymers. *Carbohydr Polym* 84(1):454–458
24. Kartel MT, Kupchik LA, Veisov BK (1999) Evaluation of pectin binding of heavy metal ions in aqueous solutions. *Chemosphere* 38(11):2591–2596
25. Dronnet VM, Renard CMGC, Axelos MAV, Thibault JF (1996) Characterisation and selectivity of divalent metal ions binding by citrus and sugar-beet pectins. *Carbohydr Polym* 30:253–263
26. Hong HJ, Ryu J, Park IS, Ryu T, Chung KS, Kim BG (2016) Investigation of the strontium (Sr(II)) adsorption of an alginate microsphere as a low-cost adsorbent for removal and recovery from seawater. *J Environ Manage* 165:263–270. doi:10.1016/j.jenvman.2015.09.040
27. Li FT, Yang H, Zhao Y, Xu R (2007) Novel modified pectin for heavy metal adsorption. *Chin Chem Lett* 18:325–328
28. Mata YN, Blázquez ML, Ballester A, González F, Muñoz JA (2009) Sugar-beet pulp pectin gels as biosorbent for heavy metals: preparation and determination of biosorption and desorption characteristics. *Chem Eng J* 150:289–301
29. Ngah WSW, Teonga LC, Hanafiha MAK (2011) Adsorption of dyes and heavy metal ions by chitosan composites: a review. *Carbohydr Polym* 83:1446–1456
30. Kumar MNVR (2000) A review of chitin and chitosan applications. *React Funct Polym* 46:1–27
31. Liu B, Wang D, Yu G, Meng X (2013) Adsorption of heavy metal ions, dyes and proteins by chitosan composites and derivatives—a review. *JOUCL* 12(3):500–508
32. Ren H, Gao Z, Wu D, Jiang J, Sun Y, Luo C (2016) Efficient Pb(II) removal using sodium alginate–carboxymethylcellulose gel beads: preparation, characterization, and adsorption mechanism. *Carbohydr Polym* 137:402–409
33. Khozhaenko E, Kovalev V, Podkorytova E, Khotimchenko M (2016) Removal of the metal ions from aqueous solutions by nanoscaled low molecular pectin isolated from seagrass *Phyllospadix iwatensis*. *Sci Total Environ* 565:913–921
34. Lakouraj MM, Mojerlou F, Zare EN (2014) Nanogel and superparamagnetic nanocomposite based on sodium alginate for sorption of heavy metal ions. *Carbohydr Polym* 106:34–41
35. Reddad Z, Gerente C, Andres Y, Le Cloirec P (2002) Adsorption of several metal ions onto a low-cost biosorbent: kinetic and equilibrium studies. *Environ Sci Technol* 36(9):2067–73
36. Crini G (2006) Non-conventional low-cost adsorbents for dye removal: a review. *Bioresour Technol* 97:1061–1085
37. Wu FC, Tseng RL, Juang RS (2001) Kinetic modeling of liquid-phase adsorption of reactive dyes and metal ions on chitosan. *Water Res* 35:613–618
38. Kumar V, Kiran K (2000) The effect of retail environment on retailer performance. *J Bus Res* 49:167–181
39. Budnyak TM, Yanovska ES, Kolodyn'ska D, Sternik D, Pylpchuk IV, Ischenko MV, Tertykh VA (2016) Preparation and properties of organomineral adsorbent obtained by sol–gel technology. *J Therm Anal Calorim* 125:1335–1351. doi:10.1007/s10973-016-5581-9
40. Asgari S, Jahanshahi M, Rahimpour A (2014) Cost-effective nanoporous agar–agar polymer/nickel powder composite particle for effective bio-products adsorption by expanded bed chromatography. *J Chromatogr A* 1361:191–202
41. Sharma G, Pathania D, Naushad M (2014) Preparation, characterization and antimicrobial activity of biopolymer based nanocomposite ion exchanger pectin zirconium(IV) selenotungstophosphate: application for removal of toxic metals. *J Ind Eng Chem* 20:4482–4490
42. Landi E, Tampieri A, Celotti G, Sprio S (2000) Desiccation behavior and mechanisms of synthetic hydroxyapatites. *J Eur Ceram Soc* 20:2377–2387
43. Gun'ko VM (2014) Composite materials: textural characteristics. *Applied Surface Sci* 307:444–454. doi:10.1016/j.apsusc.2014.04.055
44. Sing KSW, Everett DH, Haul RAW, Moscou L, Pierotti RA, Rouquerol J, Siemieniowska T (1985) Reporting physisorption data for gas/solid systems. *Pure Appl Chem* 57
45. Thommes M, Kaneko K, Neimark AV, Olivier JP, Rodriguez-Reinoso F, Rouquerol J, Sing KSW (2015) Pure Appl. Chem.; aop; IUPAC Technical Report; Physisorption of gases, with special reference to the evaluation of surface area and pore size distribution (IUPAC Technical Report). doi:10.1515/pac-2014-1117.
46. Koutsopoulos S (2002) Synthesis and characterization of hydroxyapatite crystals: a review study on the analytical methods. *J Biomed Mater Res* 62(4):600–12. doi:10.1002/jbm.10280
47. Fowler BO (1974) Infrared studies of apatites. I. Vibrational assignments for calcium, strontium, and barium hydroxyapatites utilizing isotopic substitution. *Inorg Chem* 13:194–207
48. Klee WE, Engel G (1970) Infrared spectra of the phosphate ions in various apatites. *J Inorg Nucl Chem* 32:1837–1843
49. Rey C, Shimizu M, Collins B, Glimcher MJ (1991) Resolution-enhanced Fourier-transform infrared spectroscopy study of the environment of phosphate ion in the early deposits of a solid phase of calcium phosphate in bone and enamel and their evolution with age: 2. Investigations in the 3PO₄ domain. *Calcif Tissue Int* 49:383–388
50. Gadaleta SJ, Paschalis EP, Camacho NP, Betts F, Mendelshon R, Boskey AL (1995) Fourier transform infrared spectroscopy of synthetic and biological apatites. In: Amjad Z (ed) *Mineral scale formation and inhibition*. Plenum Press, New York, pp 283–294
51. Baddiel CB, Berry EE (1966) Spectra-structure correlations in hydroxyapatite and fluorapatite. *Spectrochim Acta A* 22:1407–1416
52. Arends J, Christoffersen J, Christoffersen MR, Eckert H, Fowler BO, Heughebaert JC, Nancollas GH, Yesinowski JP, Zawacki SJ (1987) A calcium hydroxyapatite precipitated from an aqueous solution; an international multimethod analysis. *J Crystal Growth* 84:512–532
53. Nelson DGA, Featherstone JDB (1982) Preparation, analysis, and characterization of carbonated apatites. *Calcif Tissue Int* 34(Suppl 2):69–81
54. Meyer JL, Fowler BO (1982) Lattice defects in nonstoichiometric calcium hydroxyapatites. A chemical approach. *Inorg Chem* 21:3029–3035
55. Montel G (1968) Physical chemistry of phosphate with the apatite structure. *Bull Soc Chim France Spec No*:1693–1700.
56. Skwarek E, Janusz W, Sternik D (2014) Adsorption of citrate ions on hydroxyapatite synthesized by various methods. *J Radioanal Nucl Chem* 299:2027–2036
57. Yariv S (2004) The role of charcoal on DTA curves of organo-clay complexes: an overview. *Appl Clay Sci* 24:225–36
58. He H, Frost RL, Bostrom T, Yuan P, Duong L, Yang D, Xi Y, Kloprogge JT (2006) Changes in the morphology of organoclays with HDTMA+ surfactant loading. *Appl Clay Sci* 31:262–271
59. Majdan M, Pikus S, Rzączyńska Z, Iwan M, Maryuk O, Kwiatkowski R, Skrzypek H (2006) Characteristics of chabazite modified by hexadecyltrimethylammonium bromide and of its affinity toward chromates. *J Mol Struct* 791:53–60
60. Estrela dos Santos J, Dockal ER, Cavalheiro ÉTG (2005) Thermal behavior of Schiff bases from chitosan. *J Therm Anal Calorim* 79:243–248
61. López FA, Mercé ALR, Alguacil FJ, López-Delgado A (2008) A kinetic study on the thermal behaviour of chitosan. *J Therm Anal Calorim* 91(2):633–639
62. Ross AB, Hall C, Anastasakis K, Westwood A, Jones JM, Crewe RJ (2011) Influence of cation on the pyrolysis and oxidation of alginates. *J Anal Appl Pyrol* 91:344–351
63. Soares JP, Santos JE, Chierice GO, Cavalheiro ETG (2004) Thermal behavior of alginate acid and its sodium salt. *Eclat Quim* 29(2):53–56
64. Smičiklas I, Dimović S, Pleč'as I, Mitrić M (2006) Removal of Co²⁺ from aqueous solutions by hydroxyapatite. *Water Res* 40:2267–2274

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